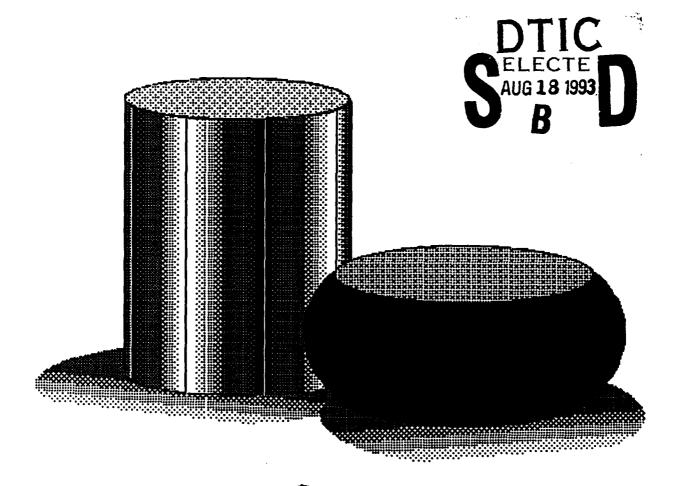
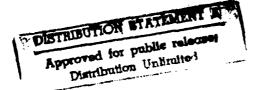
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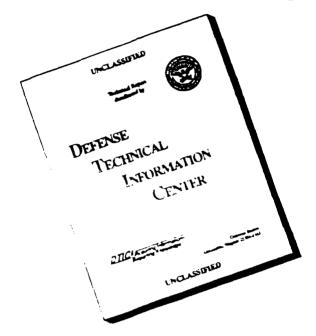
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this alloy. Workability of the	e material as a function of	of strain rate and tem	peratun	e was also determined.
Microstructural changes du	iring high temperature	deformation were	charac	terized, and selective
micrographs are presented t grain growth occurred duri	ogether with correspond	ding flow curves. I	Jynamic test on	c recrystallization and
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ATLAS OF FORMABILITY ULTIMET

by

National Center for Excellence in Metalworking Technology 1450 Scalp Avenue Johnstown, PA 15904

for

Naval Industrial Resource Support Activity Building 75-2, Naval Base Philadelphia, PA 19112-5078

March 24, 1993

This work was conducted by the National Center for Excellence in Metalworking Technology (NCEMT), operated by Concurrent Technologies Corporation (CTC), under contract to the U.S. Navy as part of the U.S. Navy Manufacturing Technology Program.

The views, opinions, and/or findings contained in this report should not be construed as an official Department of the Navy position, policy, or decision, unless so designated by other documentation.

ATLAS OF FORMABILITY

Process design and modeling engineers typically rely on materials suppliers to provide the data needed to successfully design metalforming processes. However, this information is often not sufficient or appropriate for increasingly sophisticated industrial processes and applications. In some cases, there is a total lack of mechanical and microstructural data at forming temperatures and speeds. Without proper data, engineers rely on trial-and-error methods of process development and optimization.

The National center for Excellence in Metalworking Technology (NCEMT) has begun developing an Atlas of Formability to address this problem. The objective of the Atlas is to provide the mechanical and microstructural data required to develop accurate and realistic metalforming simulations. These simulations can reduce or eliminate the need for trial-and-error approaches, thereby streamlining process development and optimization. To achieve its objective, the NCEMT uses state-of-the-art materials testing and characterization methodologies to develop workability data for advanced alloy systems. This effort is sponsored by the U.S. Navy Manufacturing Technology Program, which selects materials for the Atlas based on priorities set by the Navy. The NCEMT publishes Atlas Bulletins on a continuing basis.

ULTIMET

ABSTRACT

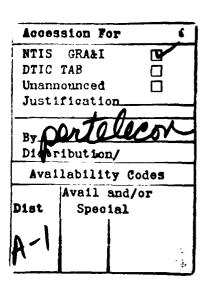
In this bulletin, flow behavior of Ultimet alloy has been determined by conducting compression tests over a wide range of temperatures (950-1200 C) and strain rates (10⁻³-20 s⁻¹). The true stress-true strain flow curves are presented for each test condition. Constitutive relations were determined from the flow behavior, and dynamic material modeling was performed. Thus, the optimum processing condition in terms of temperature and strain rate was identified as 1012 C and 10⁻³ s⁻¹. Microstructural changes during high temperature deformation were also characterized, and selected micrographs are presented together with corresponding flow curves. Dynamic recrystallization and grain growth occurred during high temperature deformation over the range of temperatures tested. This report supplies ample mechanical and microstructural data on Ultimet alloy for engineers in the field of metalforming process design.

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Ultimet

Introduction

Ultimet alloy possesses both the wear-resistant characteristics of cobalt-based Stellite alloys and the corrosion-resistant properties of nickel-based Hastelloy alloys. The alloy also has good forming and welding characteristics, but work hardening involves mechanical twinning. The understanding of mechanical and microstructural behavior during high temperature deformation is very important in forming this material. In this investigation, flow behavior of Ultimet alloy was studied by conducting compression tests over a wide range of temperatures and strain rates. Constitutive relations were determined from the flow behavior and subsequently, a dynamic material modeling for this alloy was performed. Thus, the optimum processing condition in terms of temperature and strain rate was determined. Microstructural changes during high temperature deformation were also characterized to help process design engineers select processing conditions for the desired microstructures.

Experimental Procedure

Commercially available Ultimet wrought bars in annealed condition were used in this investigation. The bars were approximately 19 mm in diameter. The typical microstructure of the asreceived material is shown in Figure 1. The chemical composition of the alloy is shown in Table 1.

Table 1. Chemical composition of Ultimet alloy.

Element	Co	Ni	Fe	Cr	Mo	W	С	N	Si	Mn	В	S	P
Wt%	bal.	8.81	3.22	26.2	5.08	2.47	.07	.092	.37	.69	.002	<.002	<.005

Cylindrical compression test specimens with a diameter of 12.7 mm and a height of 15.9 mm were machined from the bars. Isothermal compression tests were conducted in vacuum on an MTS testing machine. The test matrix was as follows:

Temperature, C (F): 950 (1742), 1000 (1832), 1050 (1922), 1100 (2012), 1125 (2057), 1150

(2102), and 1200 (2192);

Strain rate, s^{-1} : 0.001, 0.01, 0.05, 0.1, 0.5, 1, 5 and 20.

Load and stroke data were acquired during the tests by a computer and later converted to true stress-true strain curves. Immediately after the compression test, the specimens were quenched with forced helium gas in order to retain the deformed microstructure. Longitudinal sections of the specimens were examined by optical microscopy. The photomicrographs presented are electronic images and were taken from the center of the longitudinal section of the specimens.

Results

Table 2 is a list of the figures, test conditions, and the observed microstructures. True stress-true strain flow curves with selected corresponding deformed microstructure are shown from Figure 2 to Figure 57. True stress vs. strain rate was plotted in log-log scale in Figure 58 at a true strain of 0.5. The slope of the plot gives the strain rate sensitivity m, which is not constant over the ranges of temperatures and strain rates tested. Log stress vs. 1/T at the same true strain is shown in Figure 59. A processing map at this strain was developed, Figure 60. The optimum processing condition is obtained by selecting the temperature and strain rate combination which provides the maximum efficiency in the stable region. This condition is 1012 C and 10-3 s-1 for this material.

Table 2. List of figures, testing conditions and microstructural observations for Ultimet alloy.

Figure No	Temperature C (F)	Strain Rate _S -1	Microstructure	Page No
1			Optical Microscopy	
1	As received		Equiaxed grains with an average size of 51.6 μm, large cuboidal-like precipitates (ppt) and extensive twinning, some inclusions.	4
2	950 (1742)	0.001	60% dynamically recrystallized (DRX) grains with deformed (elongated) original grain boundaries still visible, necklacing at large original grain boundaries.	5
3	950 (1742)	0.01	Same as above, 50% DRX.	6
4	950 (1742)	0.05		7
5	950 (1742)	0.1	Same as above.	8
6	950 (1742)	0.5		9
7	950 (1742)	1	Same as above.	10
8	950 (1742)	5	Same as above.	11
9	950 (1742)	20	60% DRX, the new recrystallized grains have an average size of 5μm, less elongation of the original grains.	12
10	1000 (1832)	0.001	95% DRX equiaxed grains with an average size of 10µm, prior deformed grain boundaries are still visible, some ppt. and inclusions.	13
11	1000 (1832)	0.01		14
12	1000 (1832)	0.05	Same as above, but 90% DRX (prior deformed grains are well defined).	15
13	1000 (1832)	0.1		16
14	1000 (1832)	0.5	Same as above.	17
15	1000 (1832)	1	80% DRX grains with an average size of 8 µm, prior deformed grains are still visible.	18
16	1000 (1832)	5		19
17	1000 (1832)	20	Same as above, but ~95% DRX.	20
18	1050 (1922)	0.001	100% DRX equiaxed grains with an average size of 16μm, some grain growth and twinning.	21
19	1050 (1922)	0.01	Same as above except smaller grains and some prior grain boundaries.	22
20	1050 (1922)	0.05		23
21	1050 (1922)	0.1	80% DRX equiaxed grains with an average size of 10µm, deformed prior grain boundaries visible.	24
22	1050 (1922)	0.5		25
23	1050 (1922)	1	100% DRX equiaxed grains with an average size of 10µm in size, some deformed prior grain boundaries visible.	26
24	1050 (1922)	5	Same as above, but with some twinning.	27
25	1050 (1922)	20	Same as above.	28
26	1100 (2012)	0.001	100% DRX equiaxed grains with an average size of 34µm, extensive twinning.	29
27	1100 (2012)	0.01		30
28	1100 (2012)	0.05	Same as above, except that the average grain size is 25 µm.	31
29	1100 (2012)	0.1	Same as above, except that the average grain size is 21 µm.	32

30	1100 (2012)	0.5	Same as above, except that the average grain size is 20 µm.	
31	1100 (2012)	1	Same as above, except slightly smaller grains.	34
32	1100 (2012)	5		35
33	1100 (2012)	20	Same as above, except that the average grain size is 17µm.	36
34	1125 (2057)	0.001	100% DRX equiaxed grains with an average grain size of 45µm, much twinning.	37
35	1125 (2057)	0.01	Same as above, except smaller grains.	38
36	1125 (2057)	0.05		39
37	1125 (2057)	0.1	Same as above, except even smaller grains.	40
38	1125 (2057)	0.5		41
39	1125 (2057)	1	Same as above.	42
40	1125 (2057)	5	Same as above.	43
41	1125 (2057)	20	Same as above, except that the grain size is ~25µm.	44
42	1150 (2102)	0.001	100% DRX equiaxed grains with an average grain	45
42	1160 (0100)	0.01	size of 65 µm, extensive twinning.	46
43	1150 (2102)	0.01		46
44	1150 (2102)	0.05	Same as above, except that the grains are smaller.	47
45	1150 (2102)	0.1	 	48
46	1150 (2102)	0.5	Same as above.	49
47	1150 (2102)	1	Same as above, but the average grain size is 40 µm.	50
48	1150 (2102)	5		51
49	1150 (2102)	20	Same as above, except that the average grain size is 30 µm.	52
50	1200 (2192)	0.001	100 % DRX with extensive grain growth, with an average size of 79μm, much twinning.	53
51	1200 (2192)	0.01	Same as above, except that the grains are smaller.	54
52	1200 (2192)	0.05		55
53	1200 (2192)	0.1	Same as above, except that the average grain size is 40 µm.	56
54	1200 (2192)	0.5	1.7	57
55	1200 (2192)	1	Same as above, but smaller grain size.	58
56	1200 (2192)	5	Danie do decere, out oriented prunt size.	59
57	1200 (2192)	20	Same as above, except that the grains are smaller with an average size of 35 µm.	60

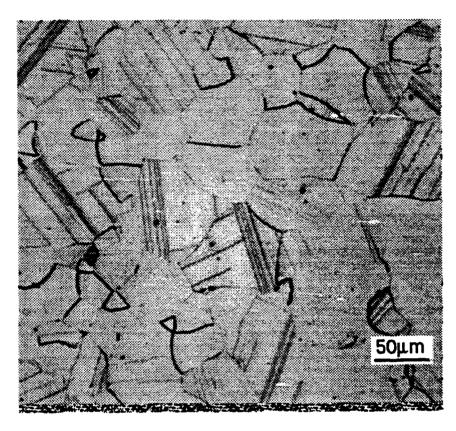
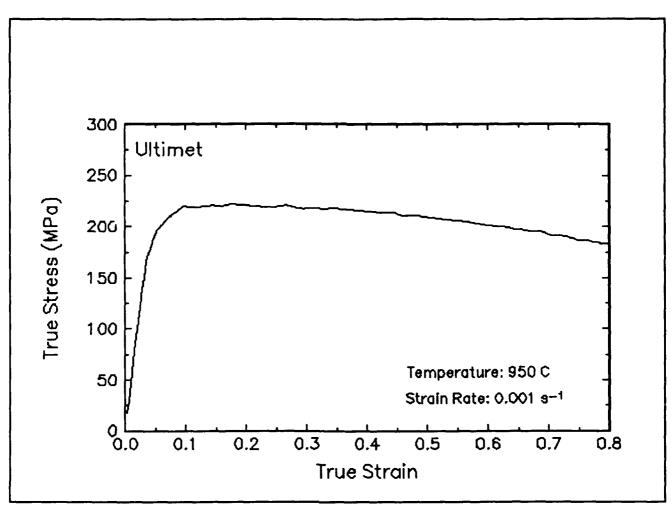


Figure 1. As-received microstructure of Ultimet



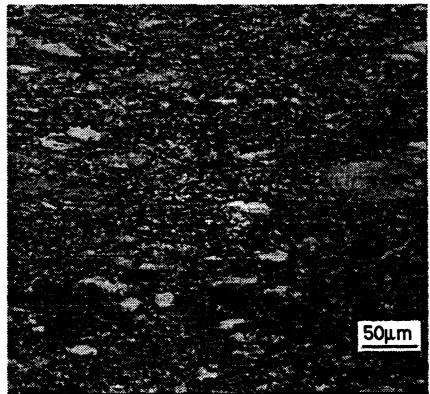
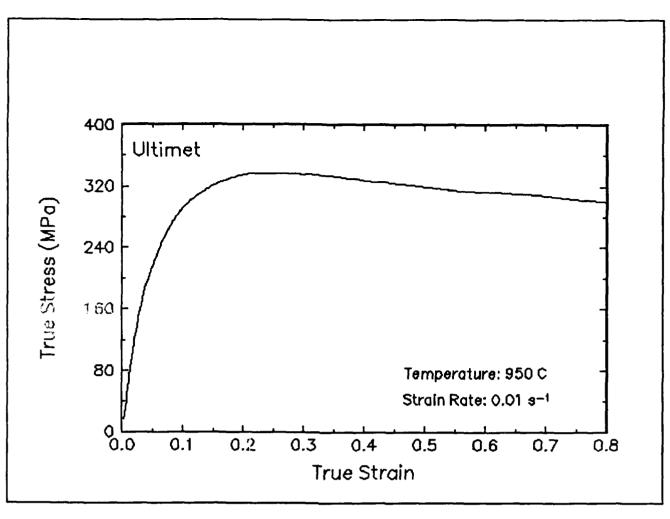


Figure 2. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 950 C and 0.001 s⁻¹.



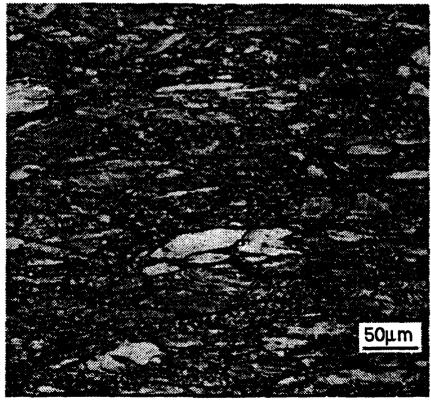


Figure 3. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 950 C and 0.01 s⁻¹.

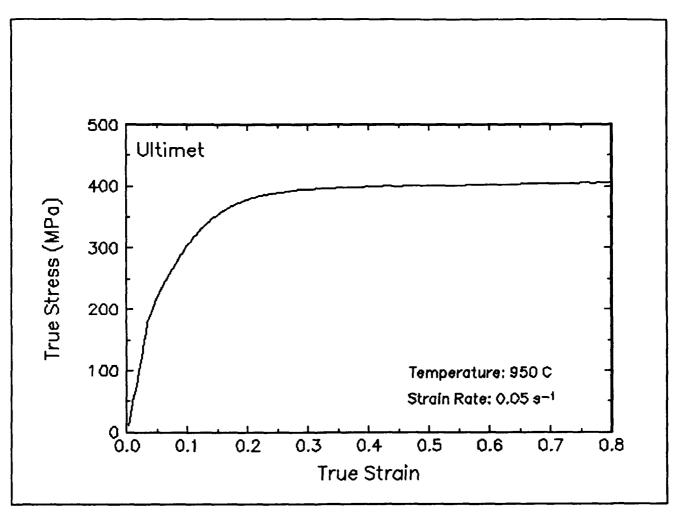
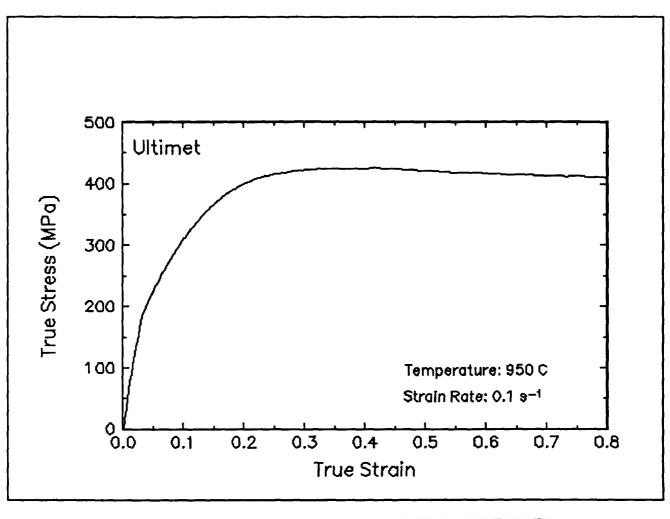


Figure 4. True stress-true strain curve, 950 C and 0.05 s⁻¹.



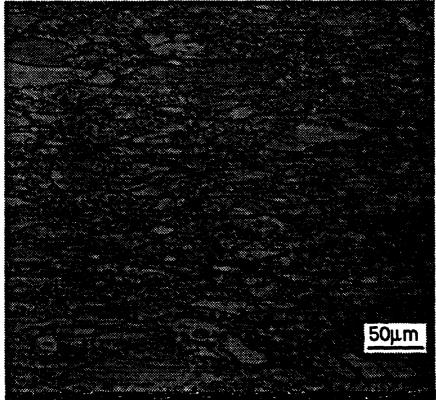


Figure 5. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 950 C and 0.1 s⁻¹.

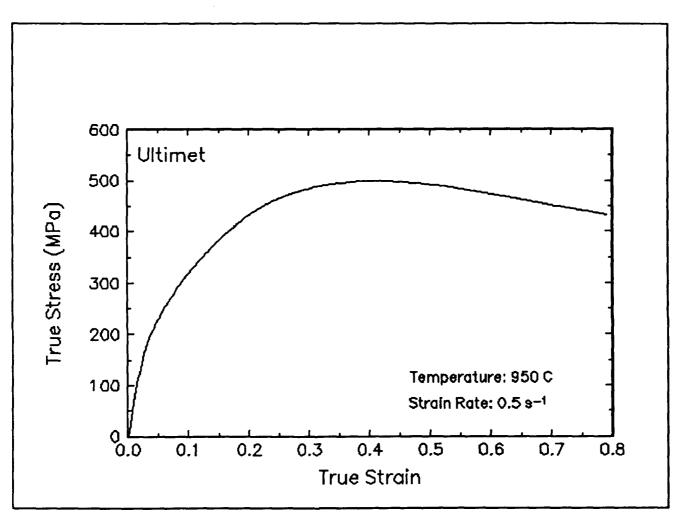
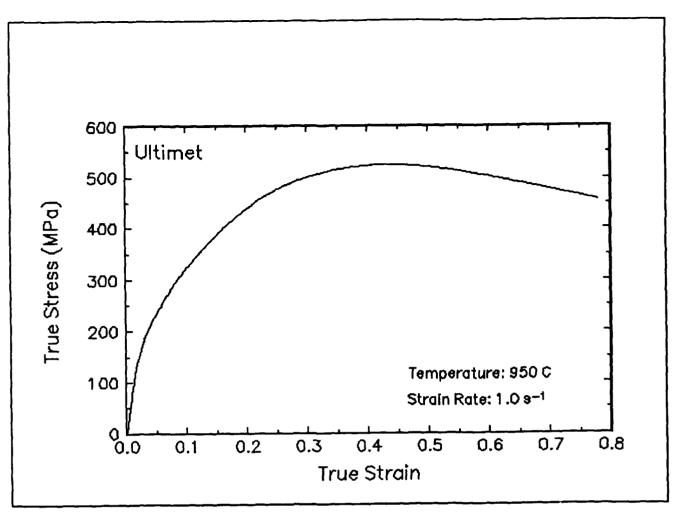


Figure 6. True stress-true strain curve, 950 C and 0.5 s⁻¹.



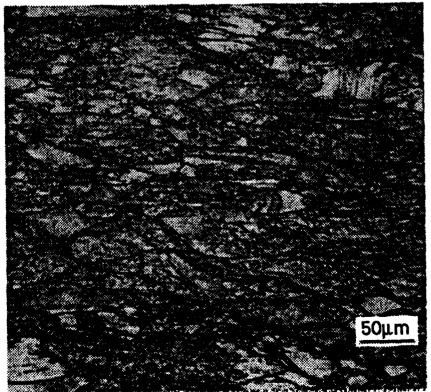
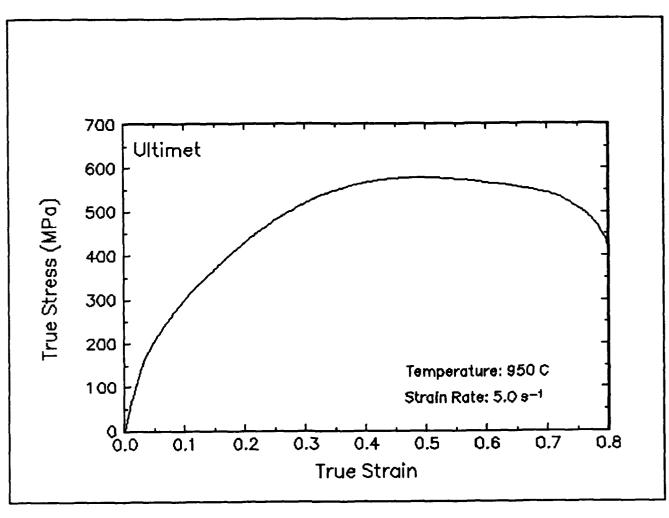


Figure 7. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 950 C and 1 s⁻¹.



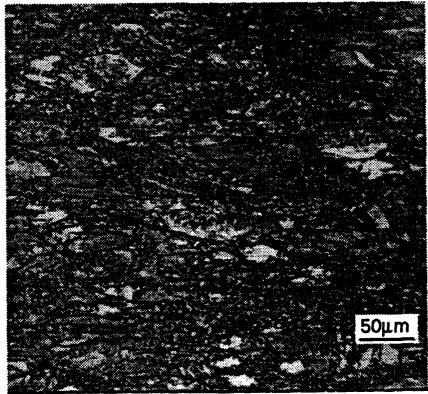
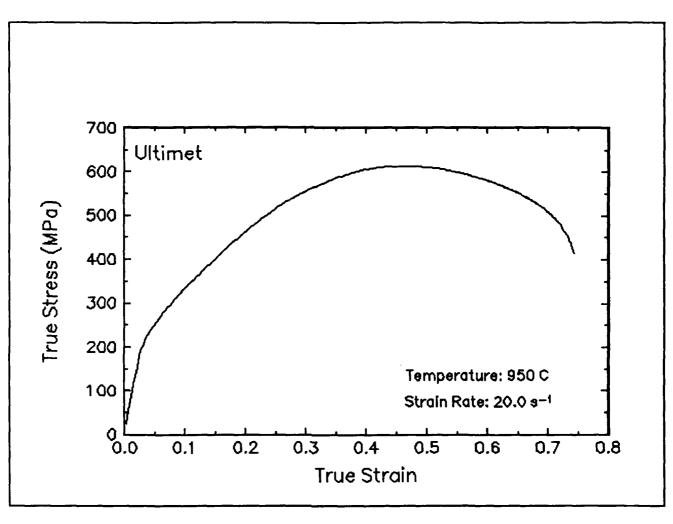


Figure 8. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 950 C and 5 s⁻¹.



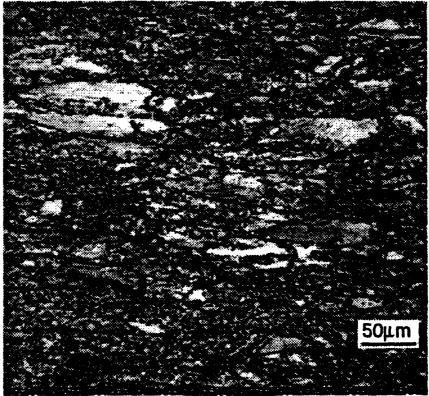
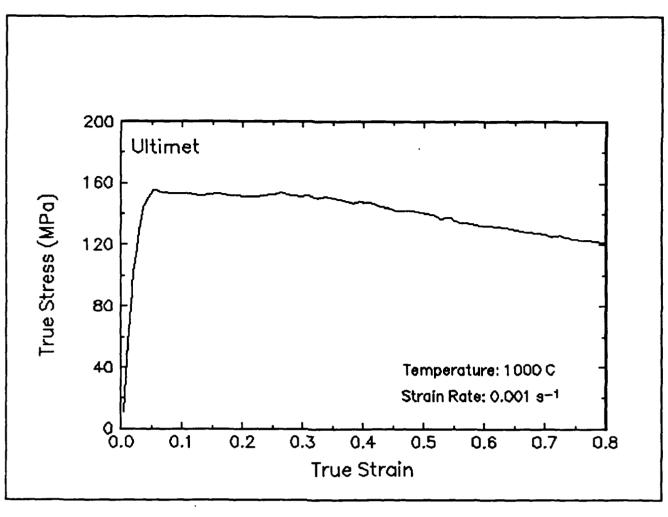


Figure 9. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 950 C and 20 s⁻¹.



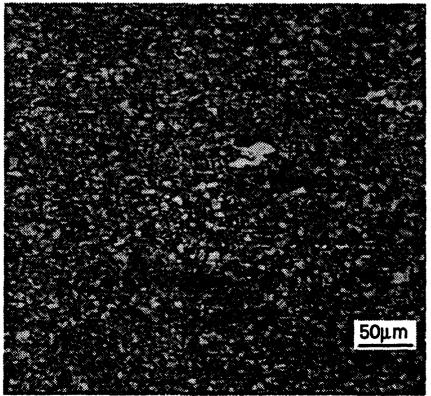


Figure 10. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1000 C and 0.001 s⁻¹.

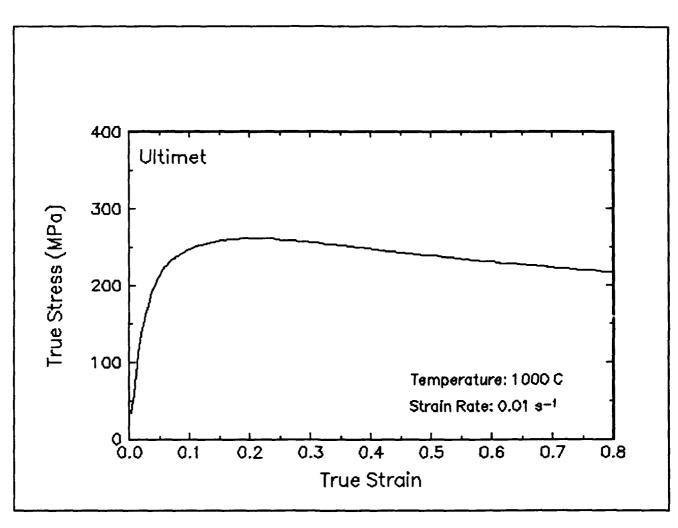
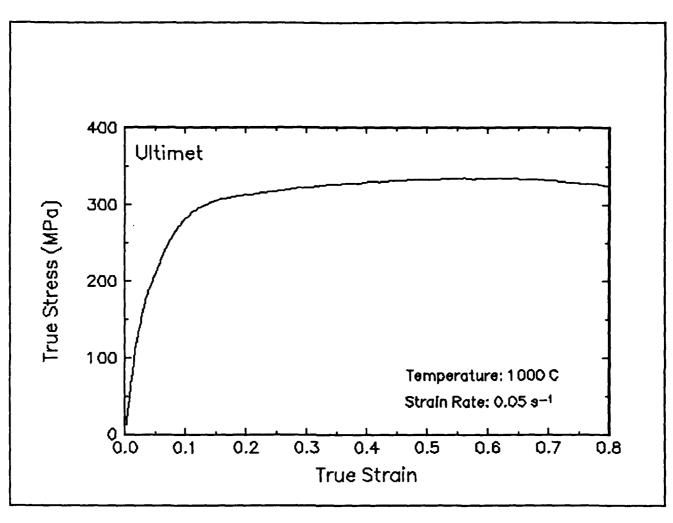


Figure 11. True stress-true strain curve, 1000 C and 0.01 s⁻¹.



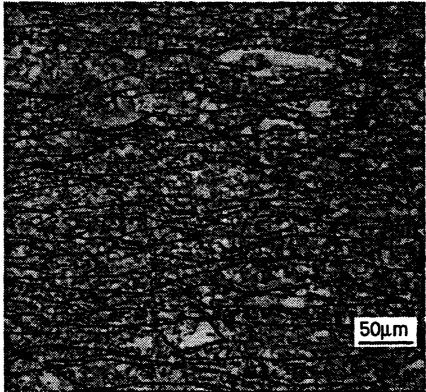


Figure 12. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1000 C and 0.05 s⁻¹.

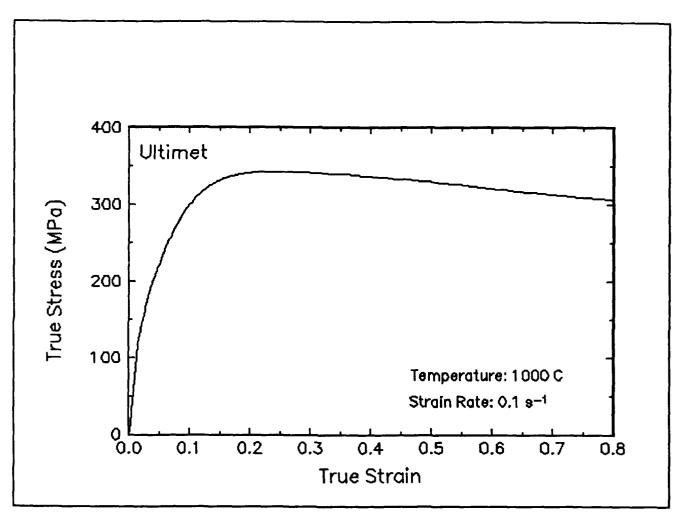
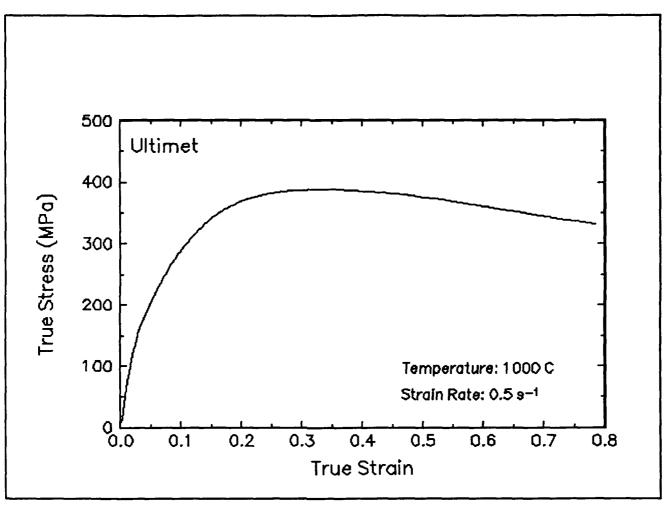


Figure 13. True stress-true strain curve, 1000 C and 0.1 s⁻¹.



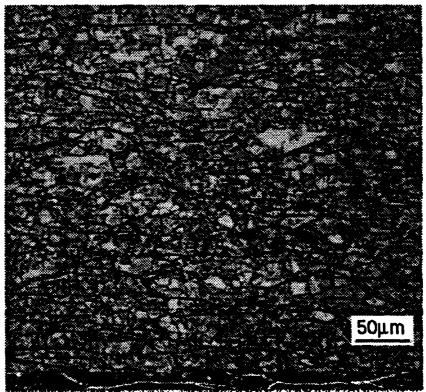
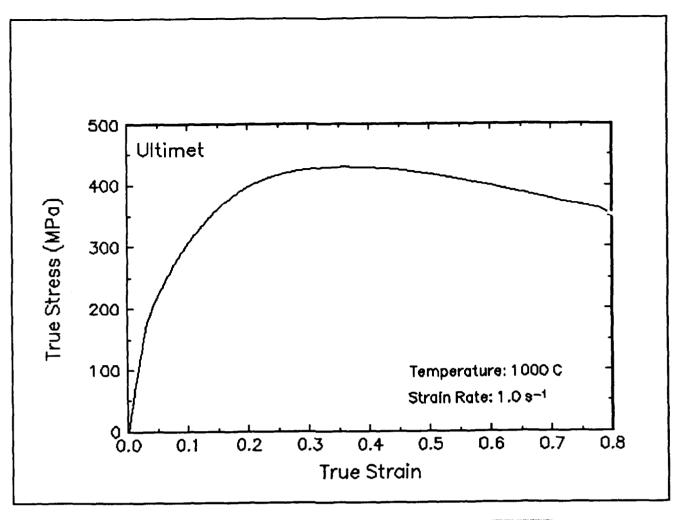


Figure 14. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1000 C and 0.5 s⁻¹.



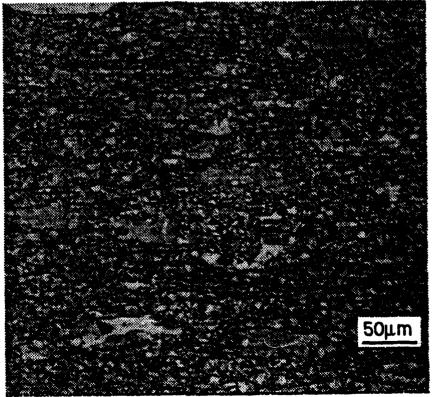


Figure 15. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1000 C and 1 s⁻¹.

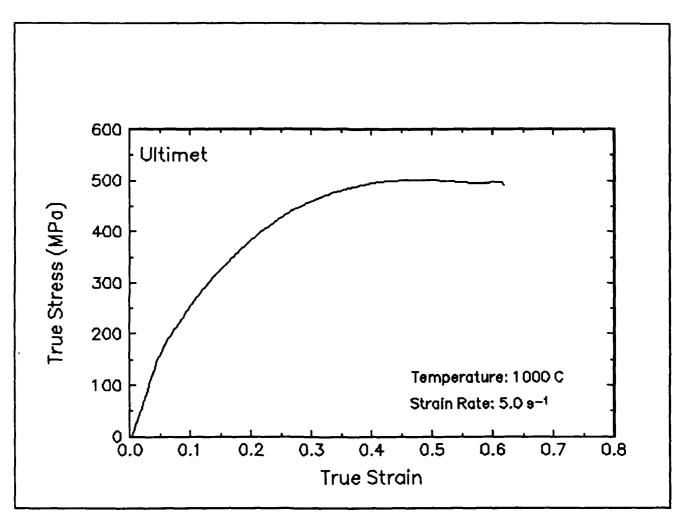
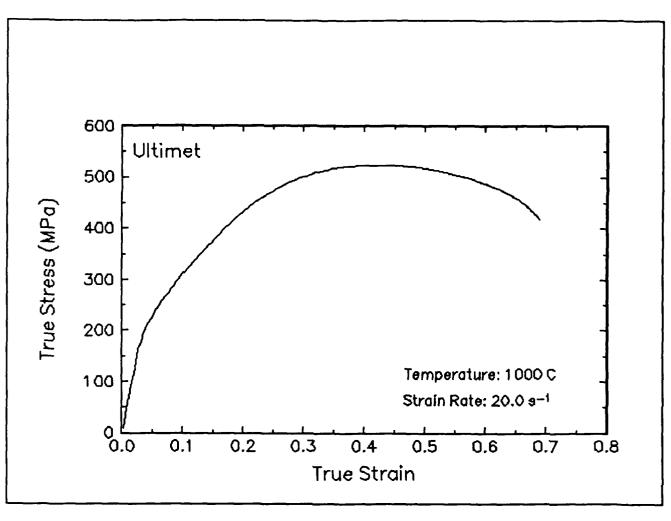


Figure 16. True stress-true strain curve, 1000 C and 5 s⁻¹.



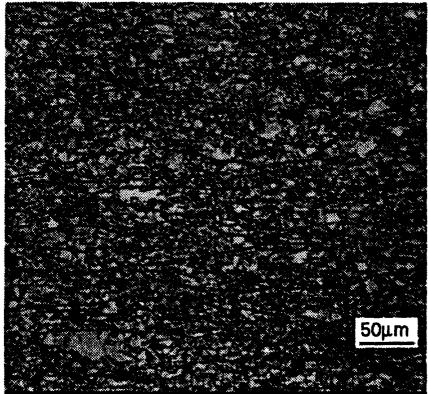
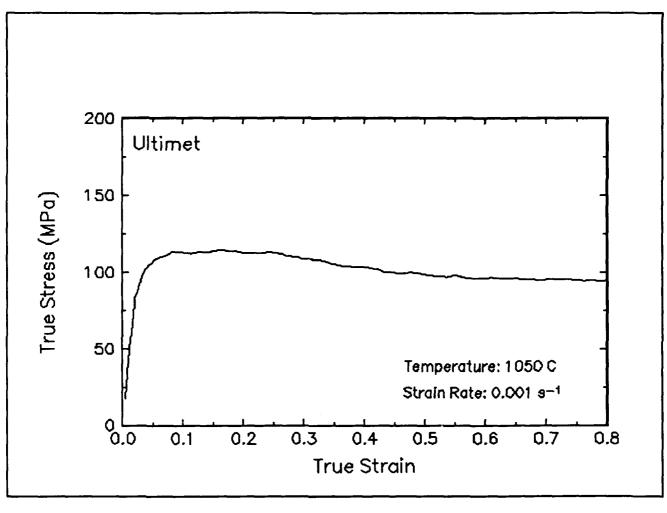


Figure 17. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1000 C and 20 s⁻¹.



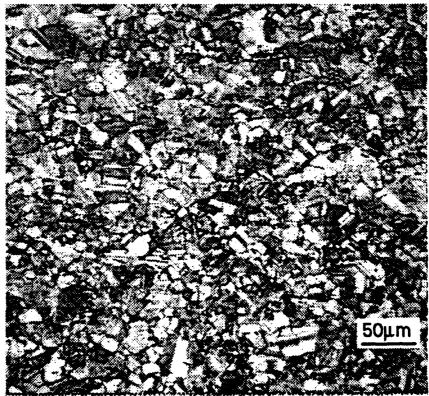
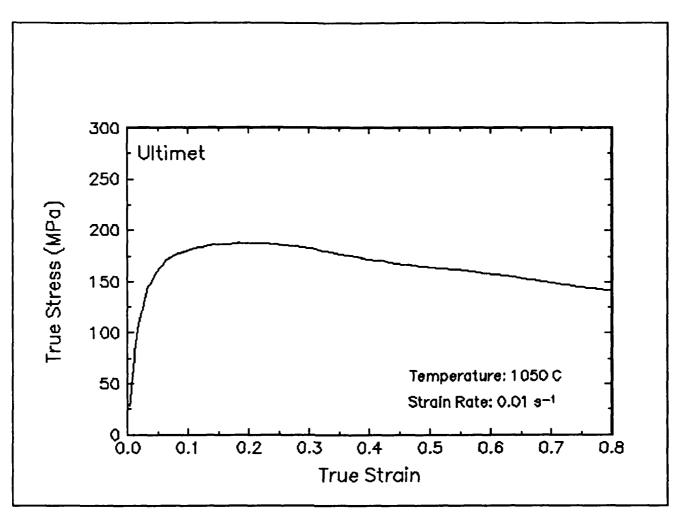


Figure 18. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1050 C and 0.001 s⁻¹.



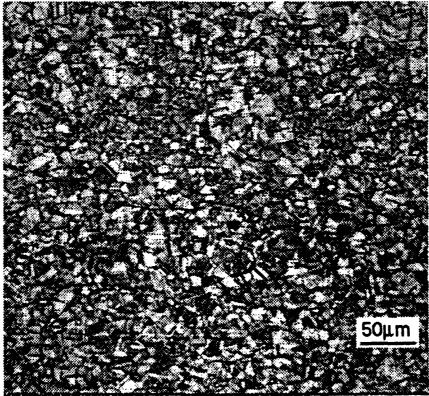


Figure 19. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1050 C and 0.01 s⁻¹.

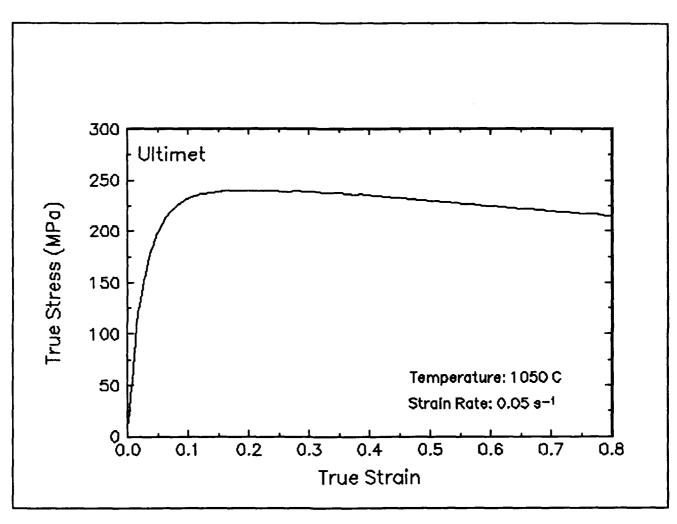
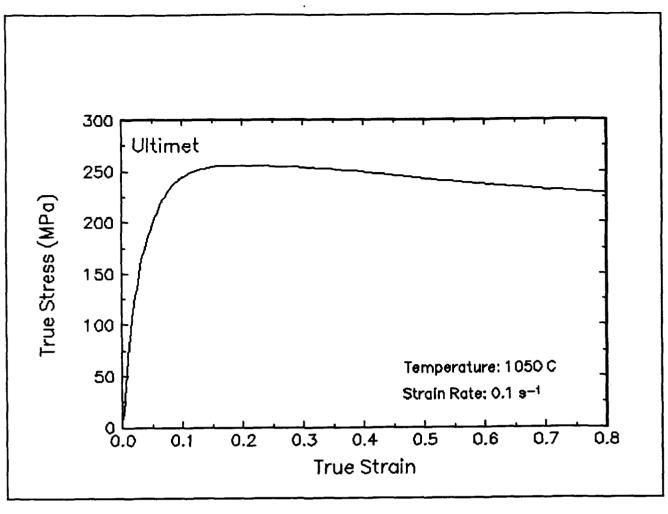


Figure 20. True stress-true strain curve, 1050 C and 0.05 s⁻¹.



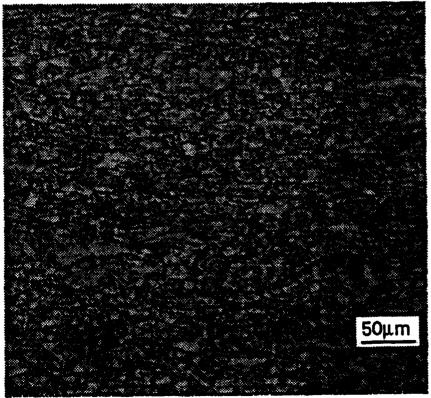


Figure 21. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1050 C and 0.1 s⁻¹.

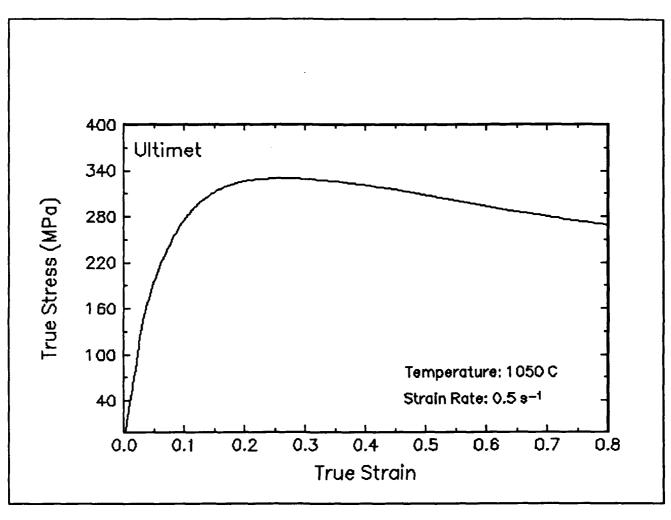
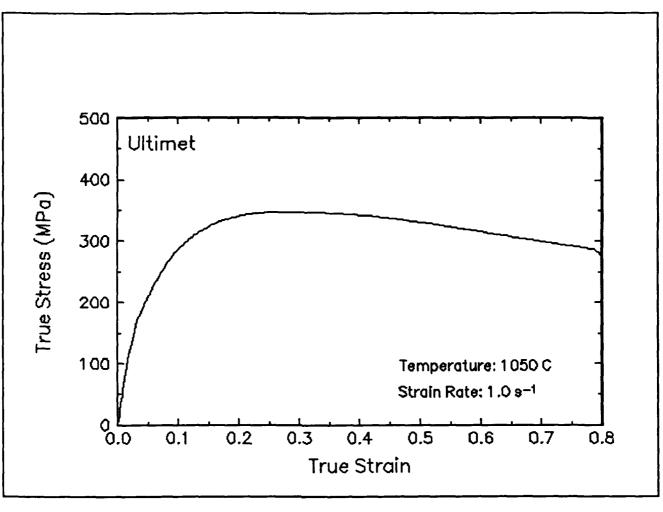


Figure 22. True stress-true strain curve, 1050 C and 0.5 s⁻¹.



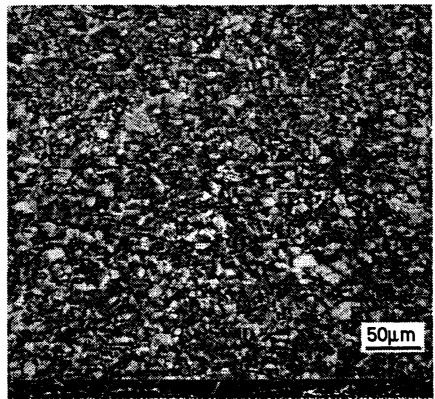
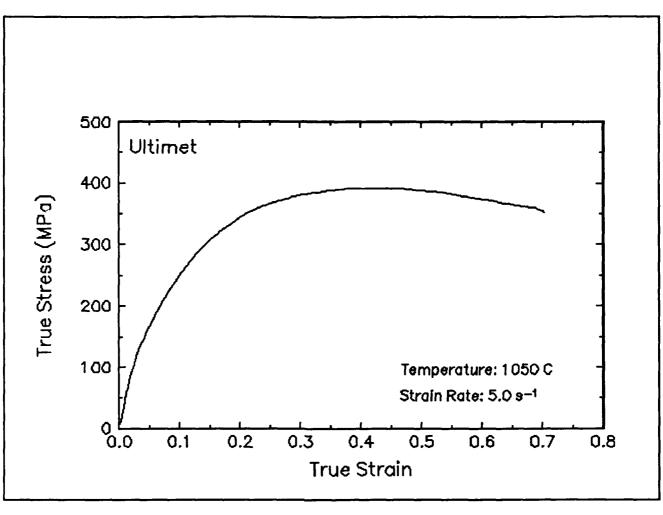


Figure 23. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1050 C and 1 s⁻¹.



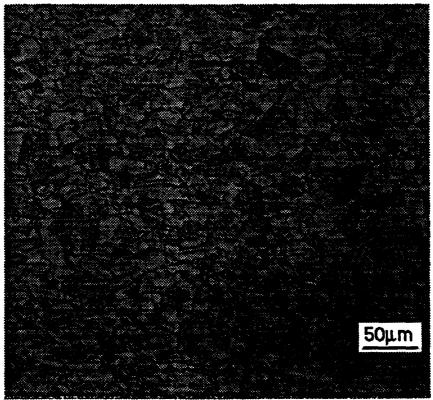
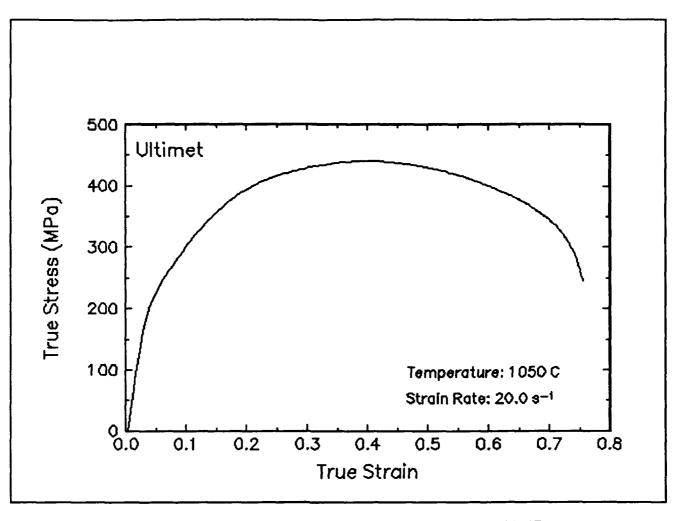


Figure 24. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1050 C and 5 s⁻¹.



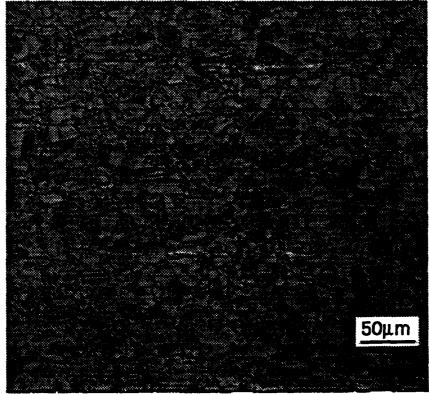
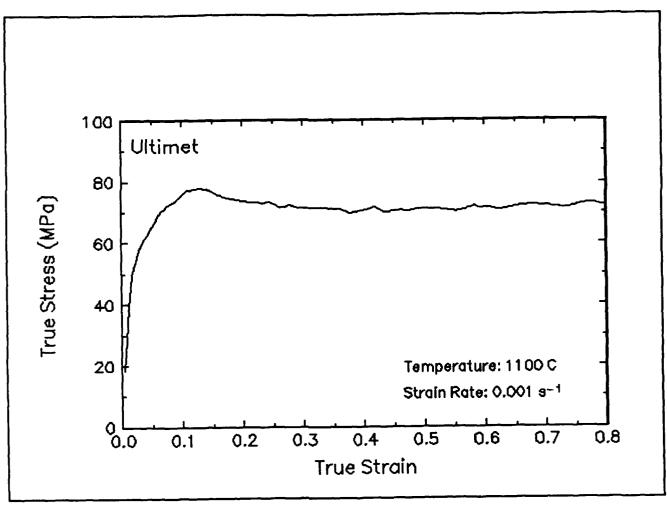


Figure 25. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1050 C and 20 s⁻¹.



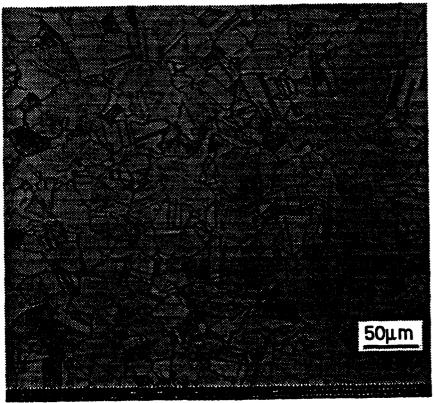


Figure 26. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1100 C and 0.001 s⁻¹.

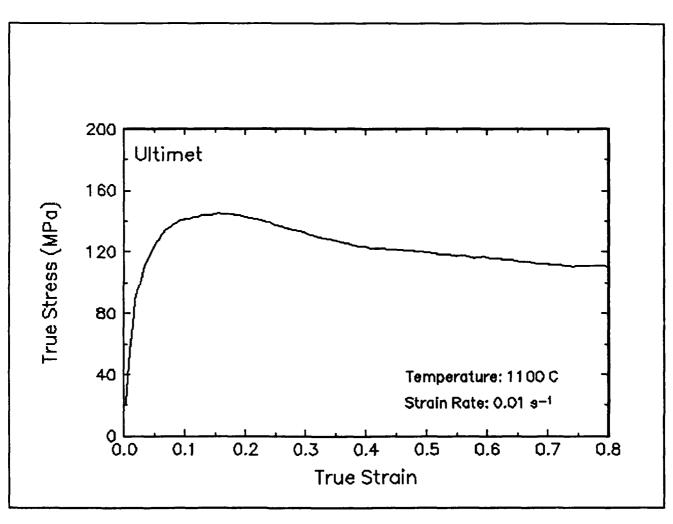
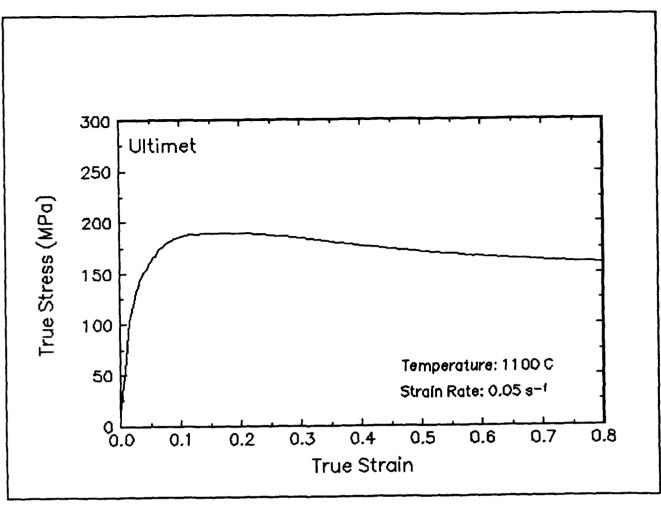


Figure 27. True stress-true strain curve, 1100 C and 0.01 s⁻¹.



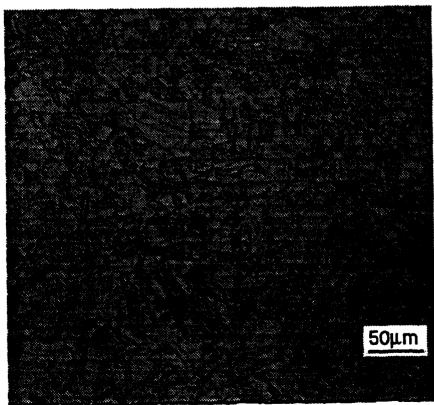
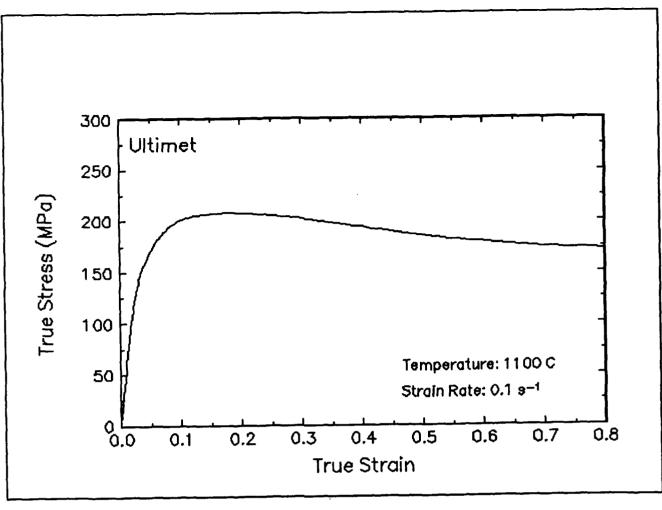


Figure 28. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1100 C and 0.05 s⁻¹.



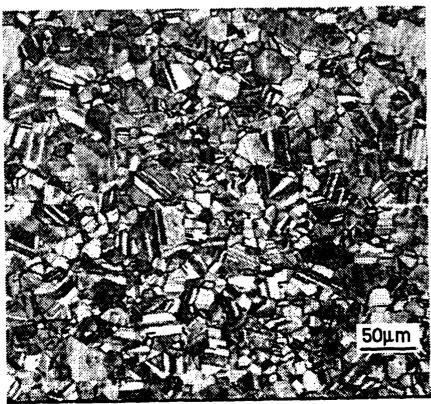
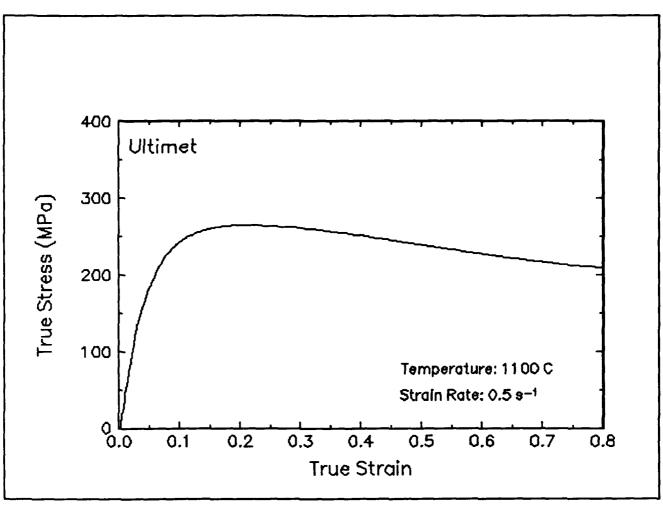


Figure 29. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1100 C and 0.1 s⁻¹.



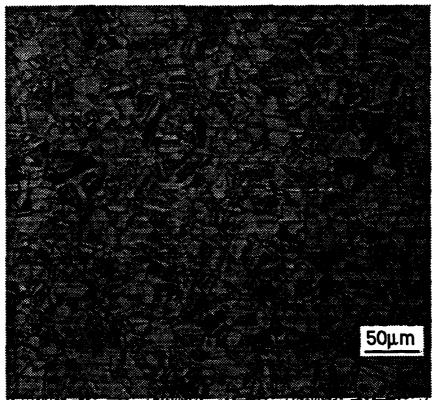
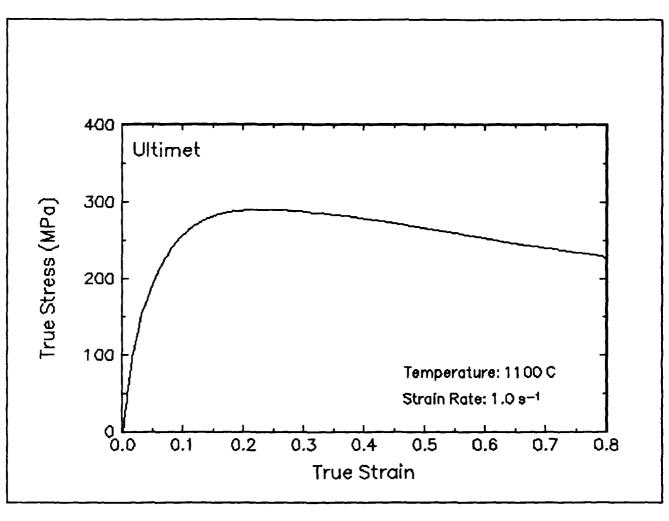


Figure 30. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1100 C and 0.5 s⁻¹.



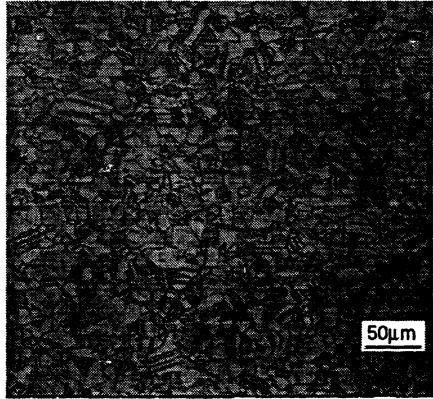


Figure 31. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1100 C and 1 s⁻¹.

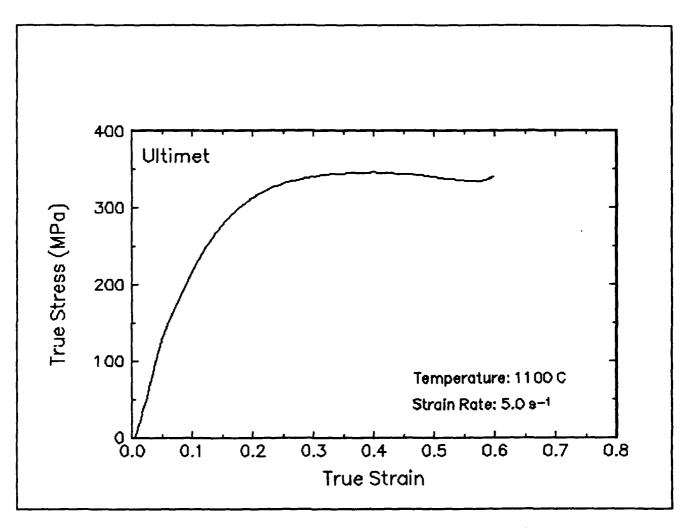
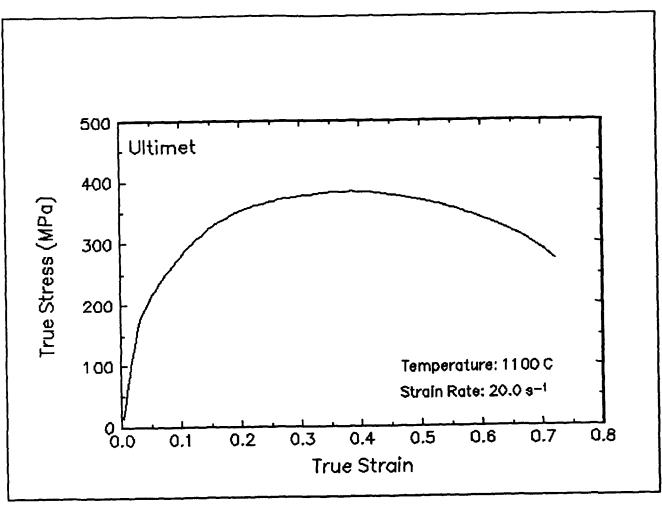


Figure 32. True stress-true strain curve, 1100 C and 5 s⁻¹.



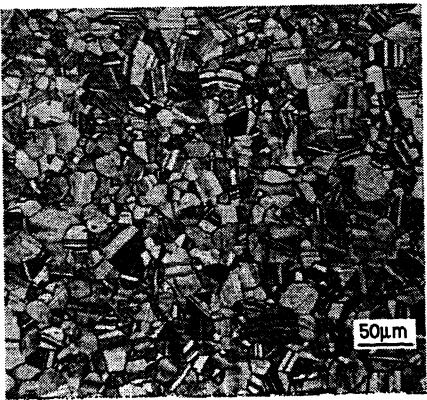
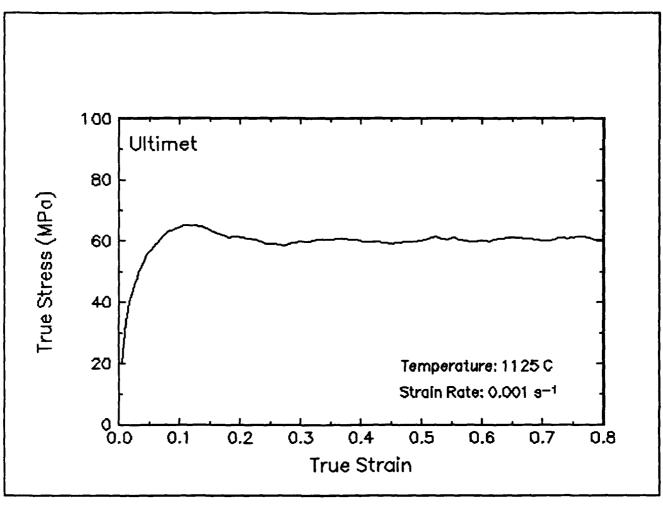


Figure 33. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1100 C and 20 s⁻¹.



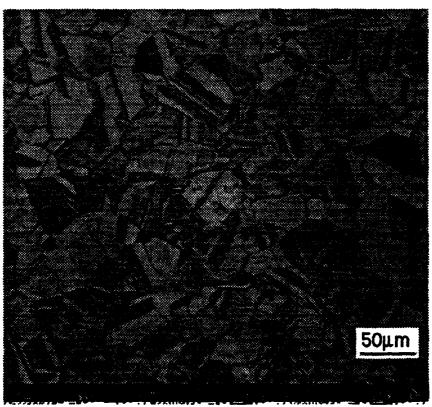
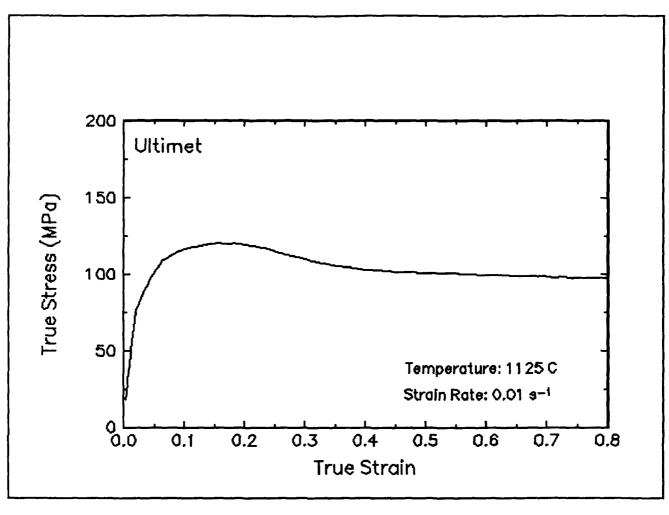


Figure 34. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1125 C and 0.001 s⁻¹.



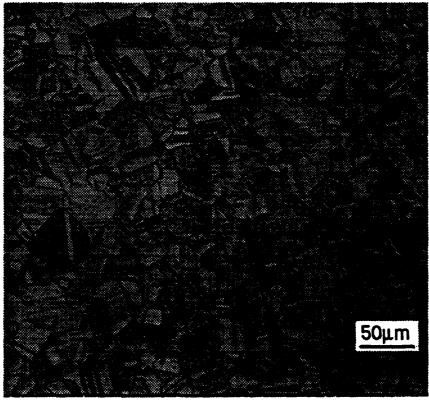


Figure 35. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1125 C and 0.01 s⁻¹.

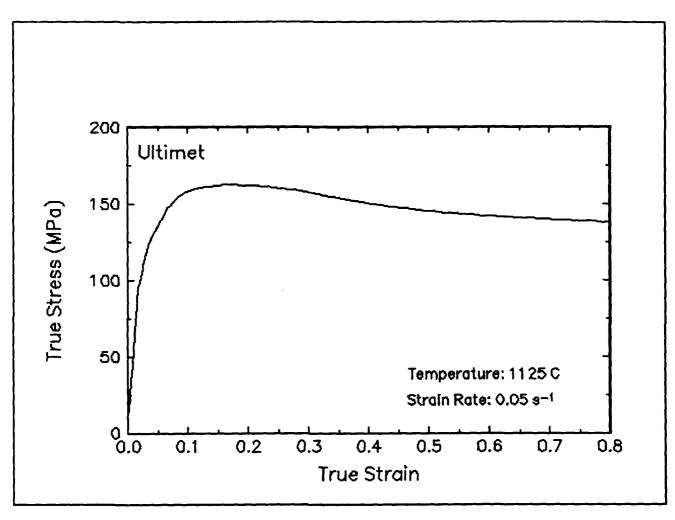
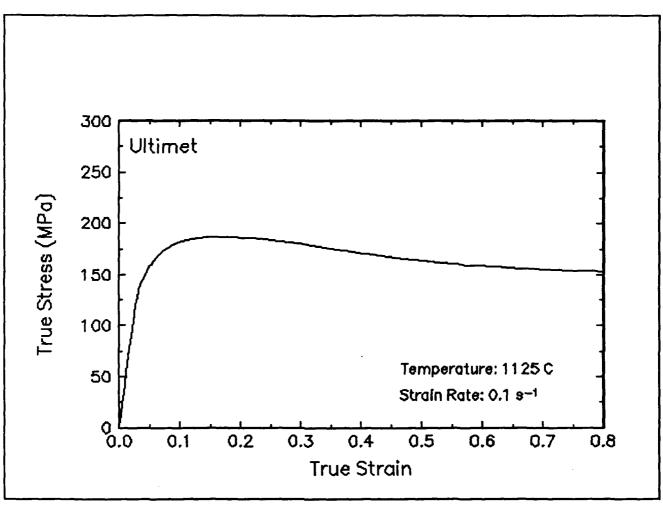


Figure 36. True stress-true strain curve, 1125 C and 0.05 s⁻¹.



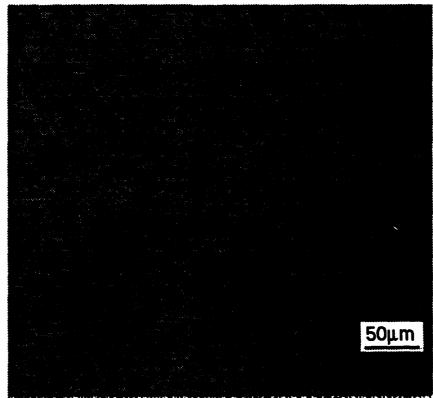


Figure 37. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1125 C and 0.1 s⁻¹.

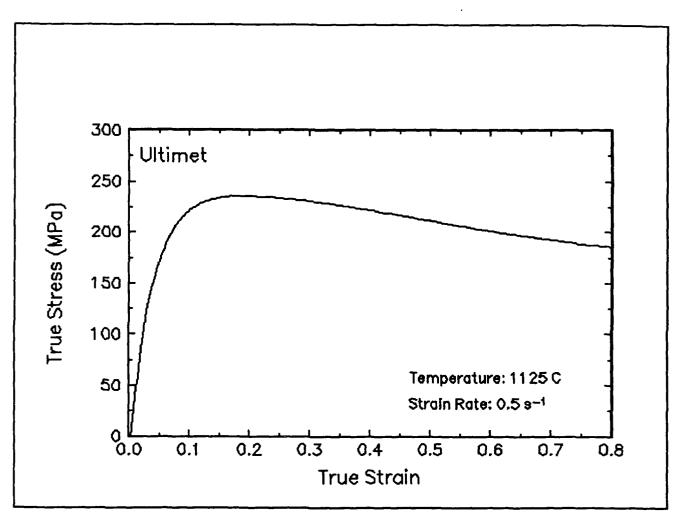
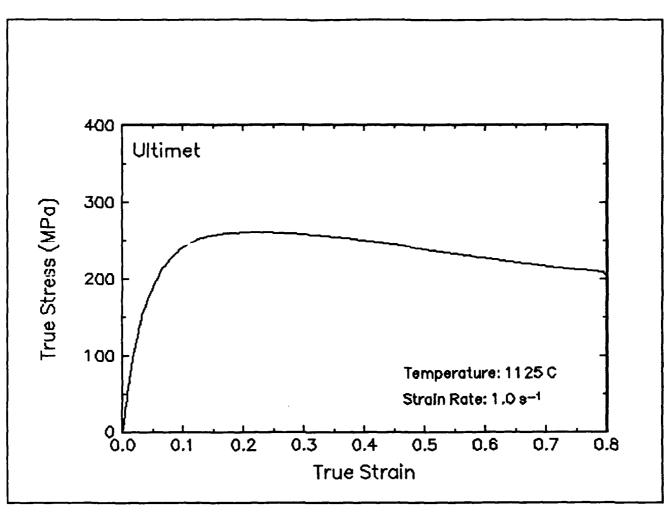


Figure 38. True stress-true strain curve, 1125 C and 0.5 s⁻¹.



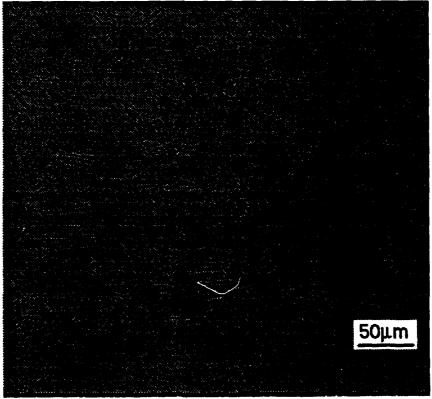
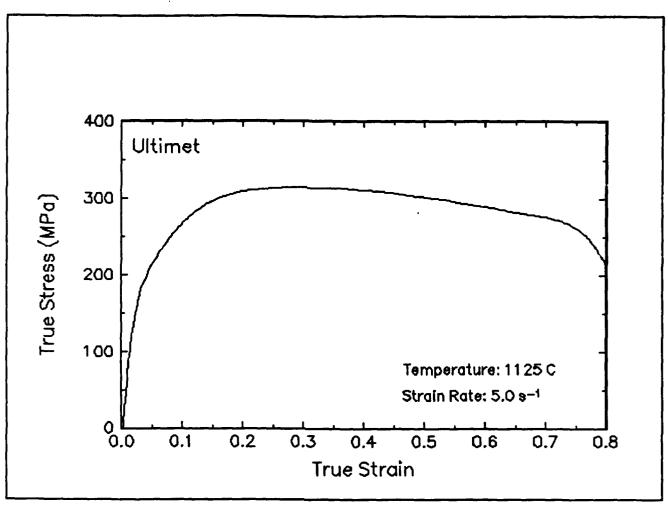


Figure 39. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1125 C and 1 s⁻¹.



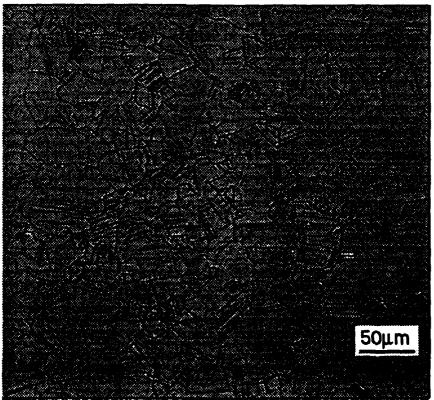
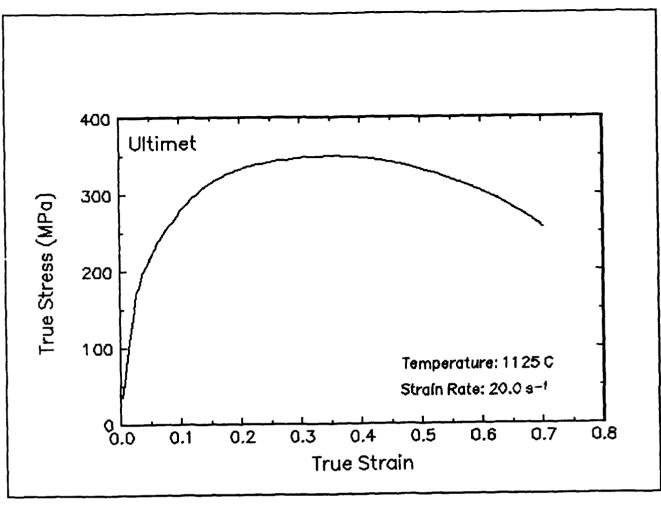


Figure 40. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1125 C and 5 s⁻¹.



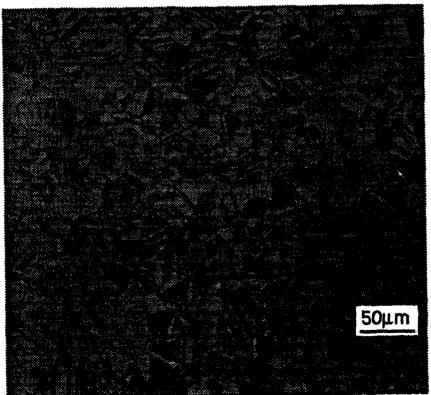
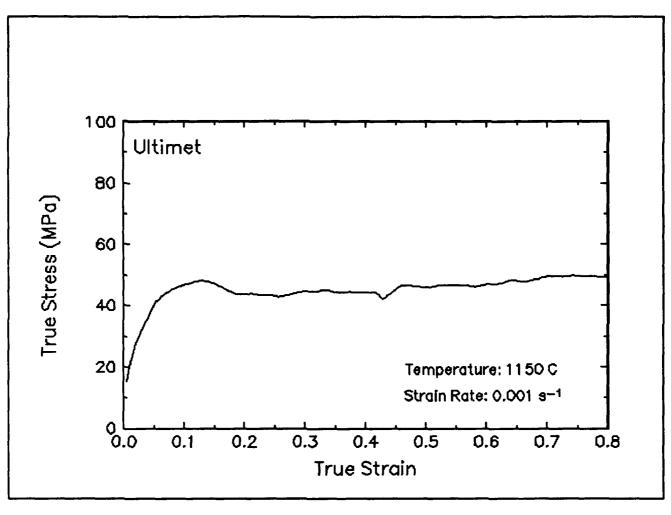


Figure 41. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1125 C and 20 s⁻¹.



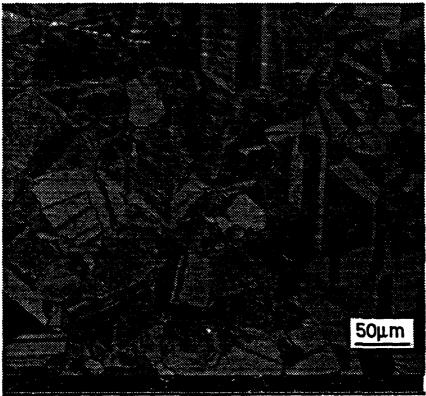


Figure 42. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1150 C and 0.001 s⁻¹.

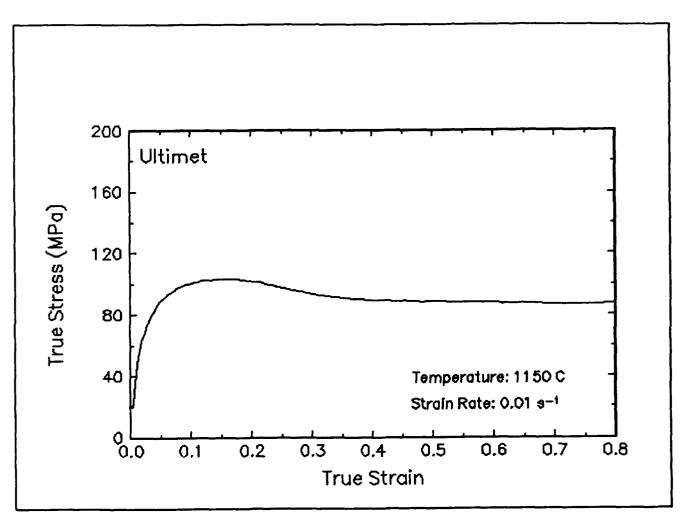
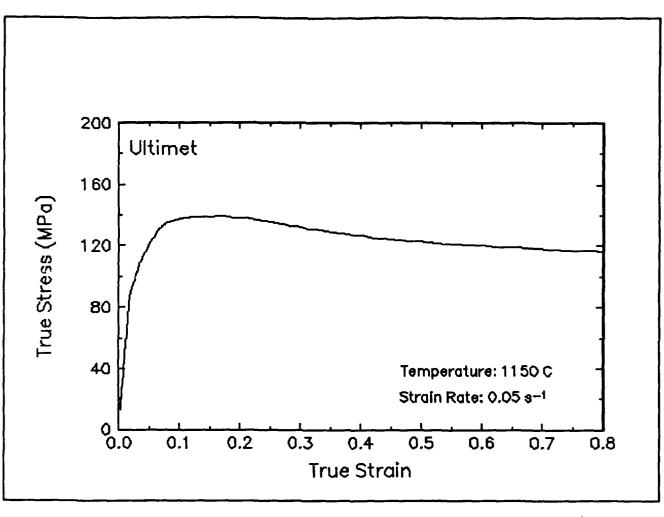


Figure 43. True stress-true strain curve, 1150 C and 0.01 s⁻¹.



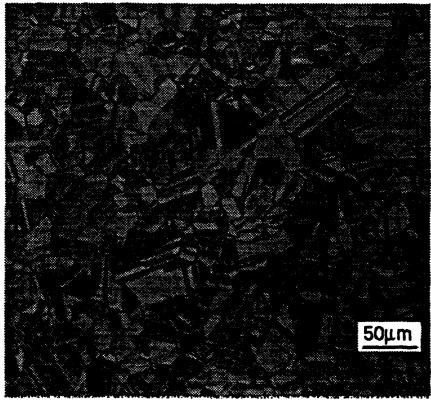


Figure 44. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1150 C and 0.05 s⁻¹.

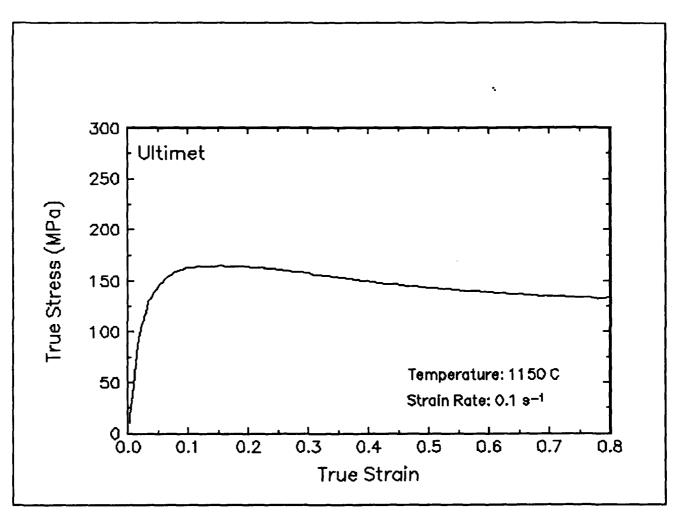
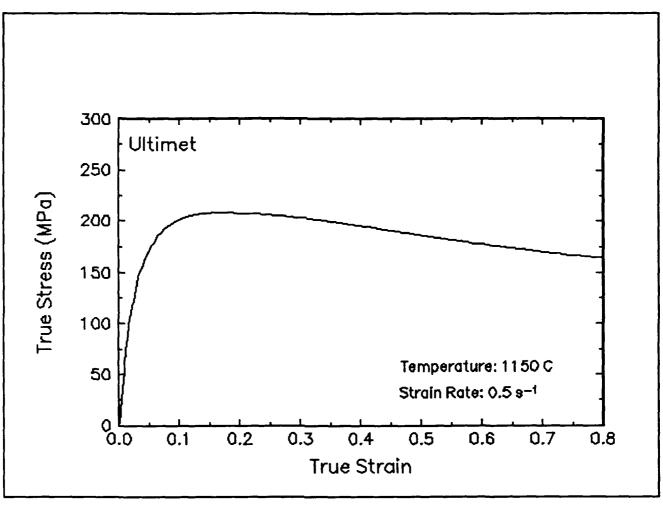


Figure 45. True stress-true strain curve, 1150 C and 0.1 s⁻¹.



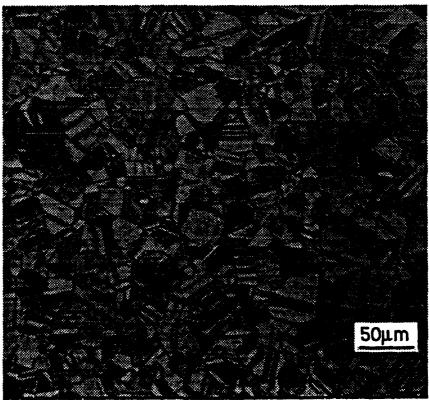
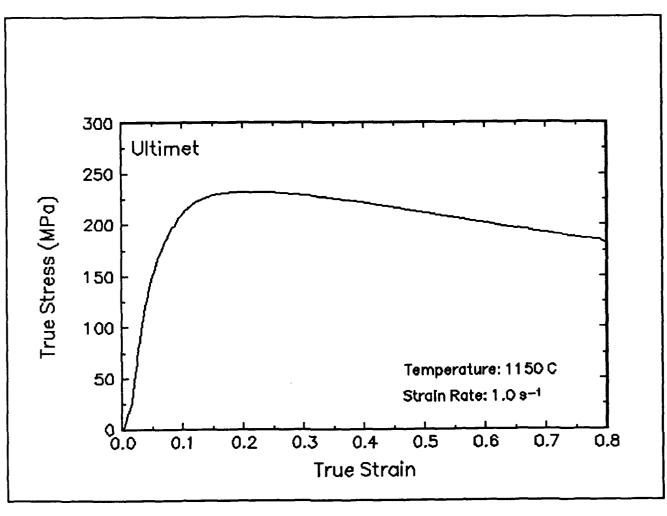


Figure 46. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1150 C and 0.5 s⁻¹.



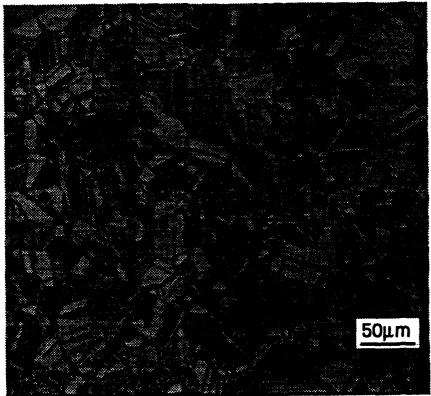


Figure 47. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1150 C and 1 s⁻¹.

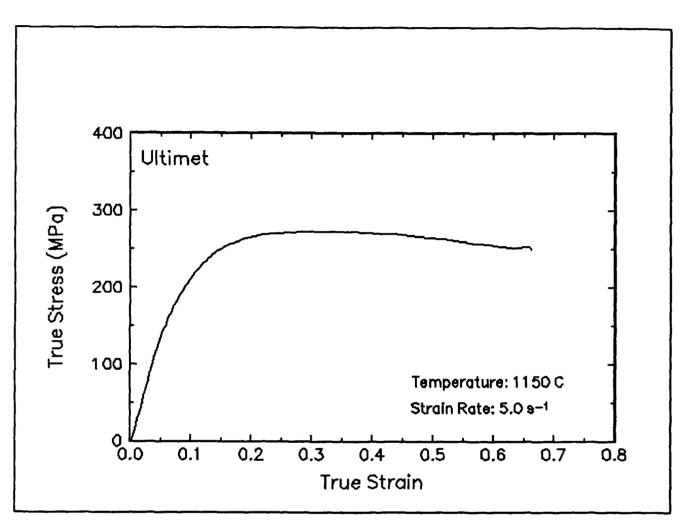
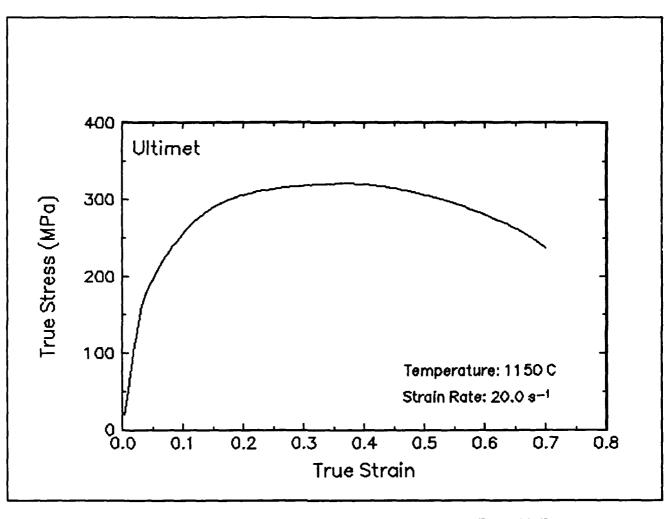


Figure 48. True stress-true strain curve, 1150 C and 5 s⁻¹.



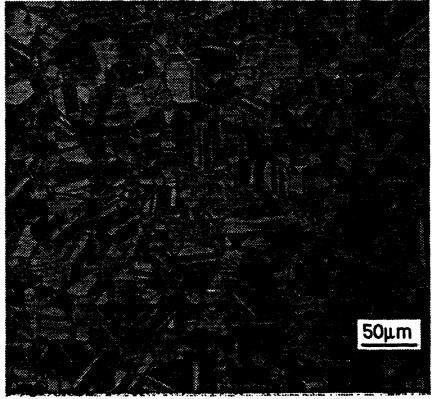
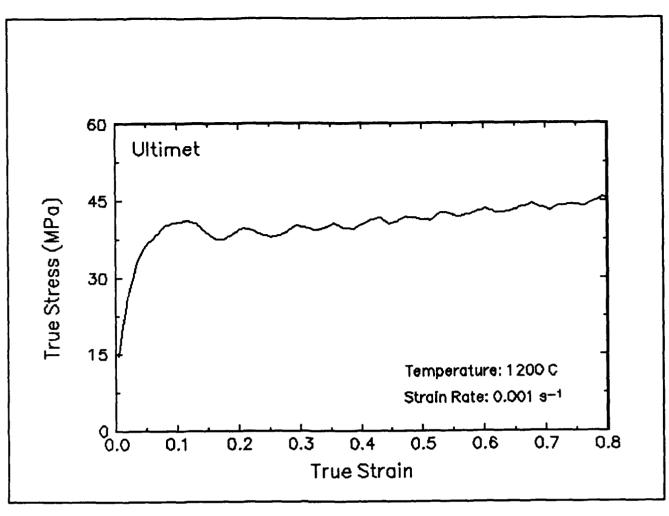


Figure 49. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1150 C and 20 s⁻¹.



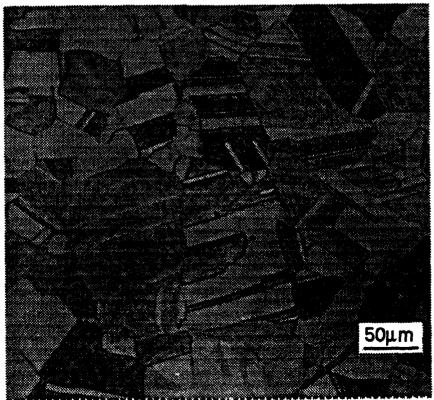


Figure 50. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1200 C and 0.001 s⁻¹.

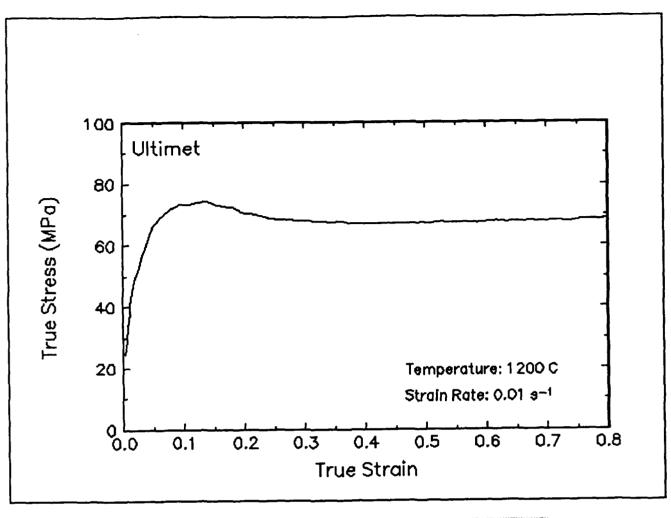




Figure 51. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1200 C and 0.01 s⁻¹.

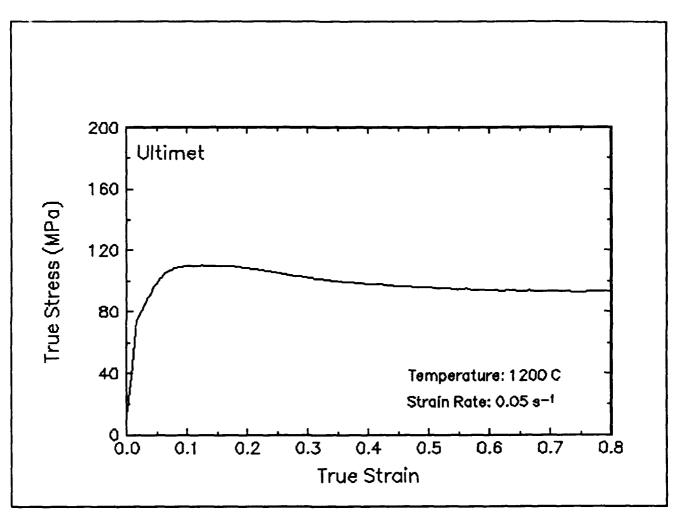
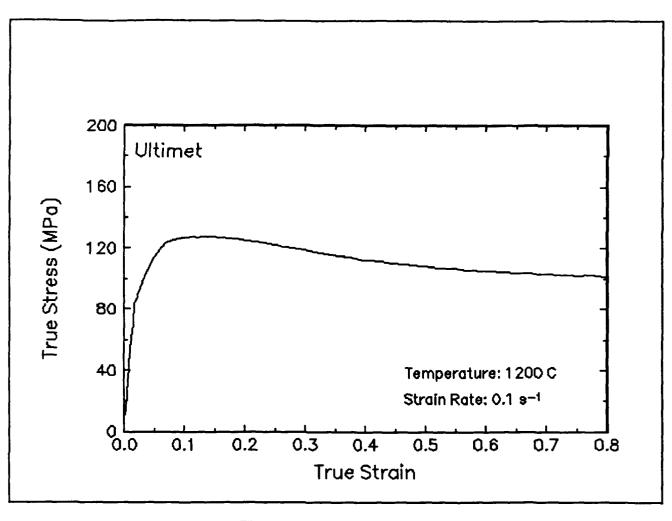


Figure 52. True stress-true strain curve, 1200 C and $0.05\ s^{-1}$.



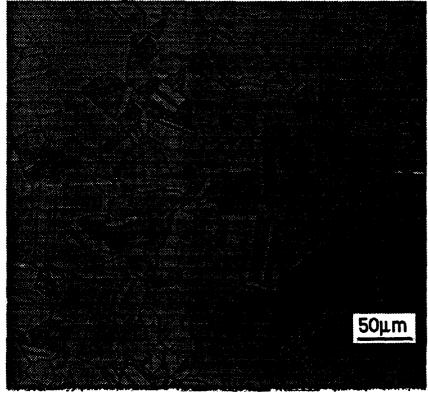


Figure 53. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1200 C and 0.1 s⁻¹.

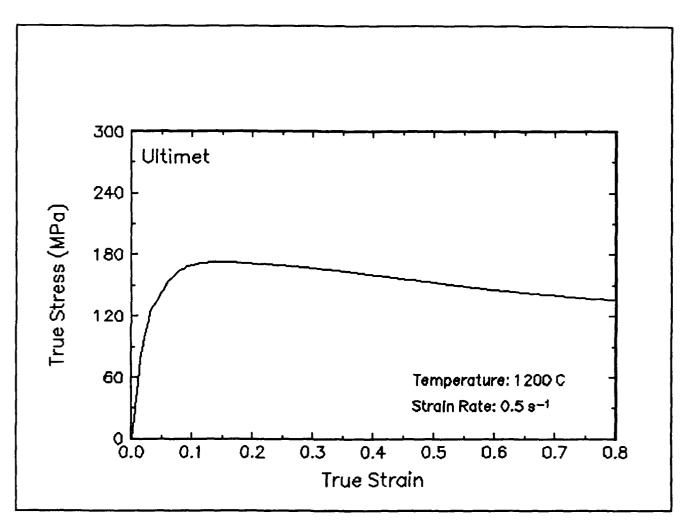
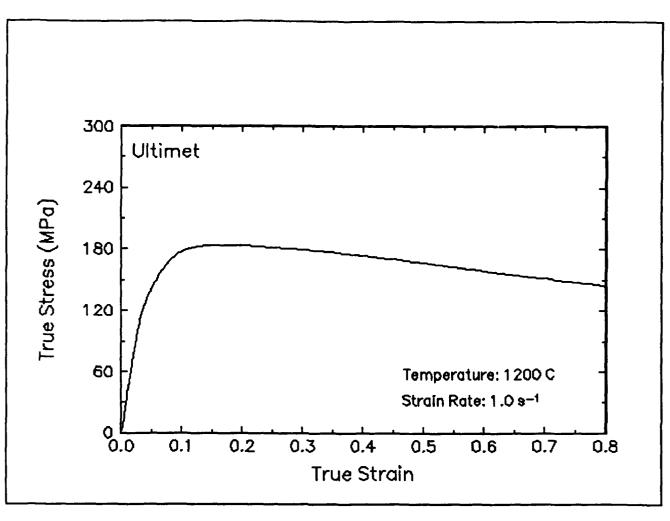


Figure 54. True stress-true strain curve, 1200 C and 0.5 s⁻¹.



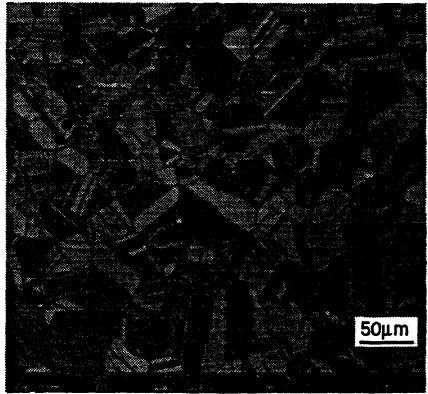
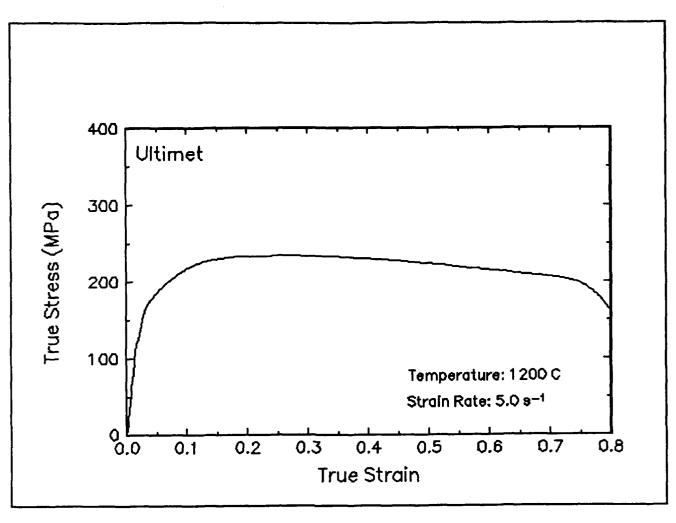


Figure 55. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1200 C and 1 s⁻¹.



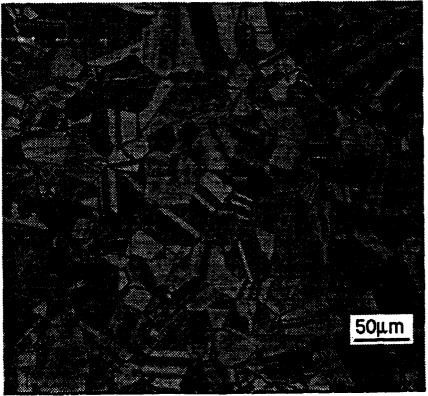
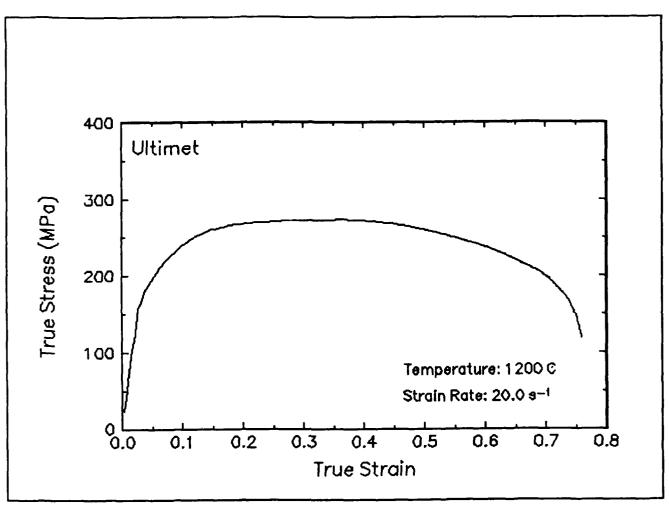


Figure 56. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1200 C and 5 s⁻¹.



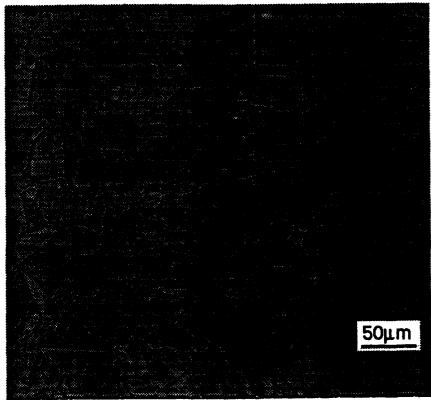


Figure 57. True stress-true strain curve and an optical micrograph from the center of the compressed sample cut through the compression axis, 1200 C and 20 s⁻¹.

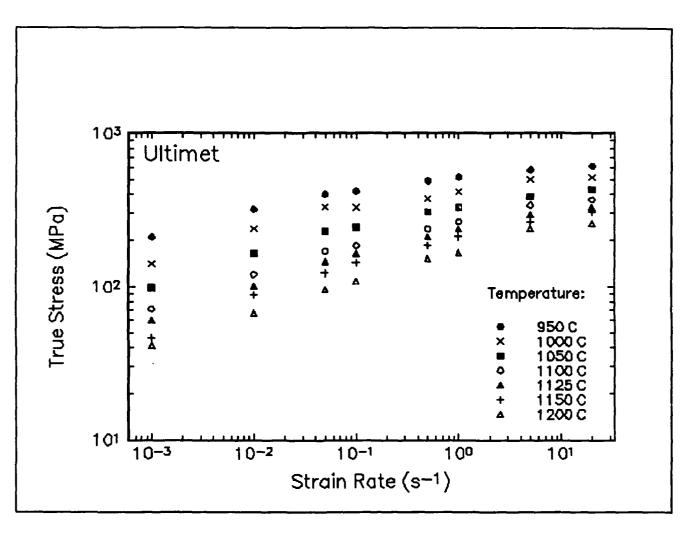


Figure 58. Effect of strain rate on stress in log-log scale at a true strain of 0.5 for Ultimet alloy.

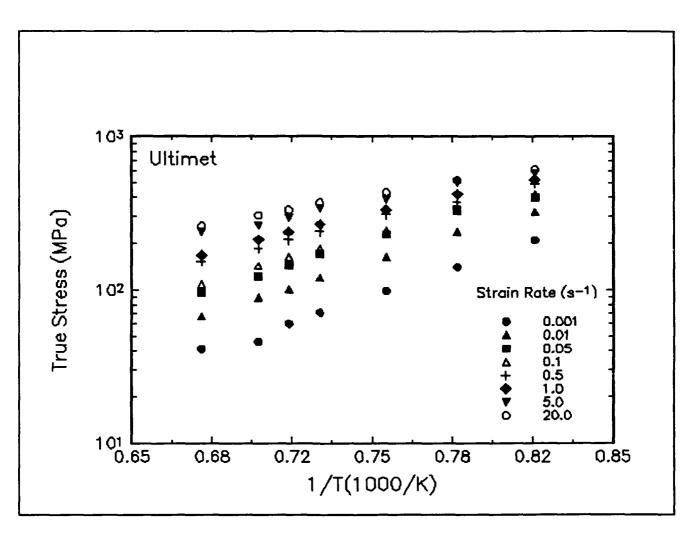


Figure 59. Effect of temperature on stress at a true strain of 0.5 for Ultimet alloy.

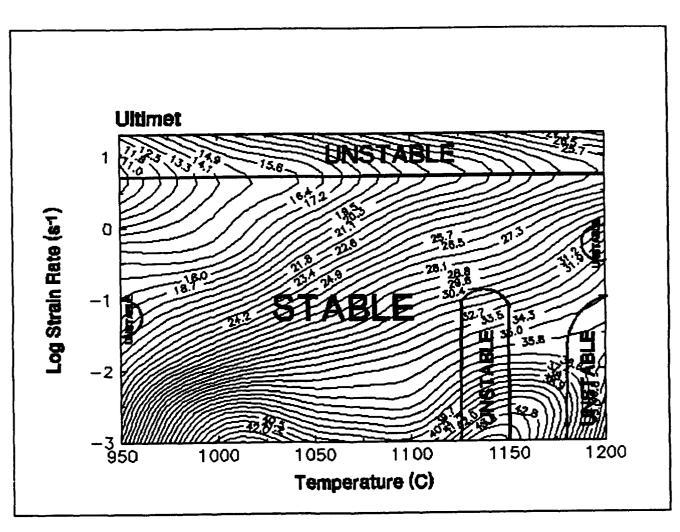


Figure 60. Processing map of Ultimet alloy at a true strain of 0.5.

Summary

Compression tests were performed on a commercially available Ultimet alloy over a wide range of temperatures and strain rates. The experimental conditions in this work were representative of those in metalforming practice. From the true stress-true strain curves, the flow behavior was characterized and a processing map, indicating the optimum processing condition, was generated. This condition is 1012 C and 10⁻³ s⁻¹.

The deformed microstructures were characterized from the quenched specimens by optical microscopy and were presented for selected test conditions together with stress-strain curves.

Implementation of Data Provided by the Atlas of Formability

The Atlas of Formability program characterizes the flow behavior of engineering materials in the temperature and strain rate regime commonly used by metalforming industry. The data are valuable in design and problem solving in metalforming processes. Microstructural changes during deformation are also provided in the Bulletin. This information helps the design engineer to select processing parameters leading to the desired microstructure.

The data can also be used to construct processing maps using the dynamic material modeling approach to determine stable and unstable regions in terms of temperature and strain rate. The temperature and strain rate combination at the highest efficiency in the stable region provides the optimum processing condition. In some metalforming processes, such as forging, the strain rate varies within the workpiece. Finite Element Analysis of the process can ensure that strain rates and processing temperatures throughout the workpiece fall into the stable regions in the processing map. Furthermore, FEM analysis based on the data from the Atlas of Formability can be coupled with fracture criteria to predict defect formation in metalforming processes.

Concurrent Technologies Corporation uses the *Atlas of Formability* data to design metalforming processes, including dynamic material modeling, FEM analysis, and defect prediction. *CTC* can apply its expertise to develop and optimize metalforming processes. For more information, please contact Dr. Prabir K. Chaudhury, Manager, Forming Department, by calling (814) 269-2594.